

4 **Influence of ionic strength on the flexibility of alginate studied by size** 5 **exclusion chromatography** 6

7 F. Guillermo Díaz Baños^{a*}, Ana I. Díez Peña^a, J. Ginés Hernández Cifre^a, M. Carmen López
8 Martínez^a, Alvaro Ortega Retuerta^b, José García de la Torre^a
9

10 ^aDepartament of Physical Chemistry, Faculty of Chemistry, Regional Campus of International Excellence
11 "Campus Mare Nostrum", University of Murcia, 30071 Murcia, Spain

12 ^b Estación experimental del Zaidín, CSIC, 18008 Granada, Spain
13

14 *Corresponding author: E-mail: fgb@um.es. Telephone: +34 868887394.
15
16
17
18
19
20

21 **Keywords:** alginate, $[\eta]$ -M relationship, flexibility, persistence length, wormlike, HYDFIT
22

23 **Abstract**

24 SEC measurements of the $[\eta]$ -M relationship for alginate from *M. pyrifera* and the wormlike model can be
25 used to characterize flexibility through two independent treatments (Bohdanecky's equations and HYDFIT
26 program), both providing the same results. Two different assumptions concerning mass per unit of length
27 lead to different conclusions. First: persistence length decreases with ionic strength (the intrinsic component
28 of the persistence length is 11.3 nm and the electrostatic component is 6 nm when ionic strength is 0.01).
29 Second: persistence length is independent of ionic strength (12 nm). Either of these options shows that the
30 wormlike model in itself is not sufficient to explain flexibility over the whole range of chain lengths for these
31 polyelectrolytes. A plausible explanation could be the presence of a combination of short-range and long-
32 range screening effects of the ions of the solutions. This would also explain some data found in the literature
33 regarding alginate flexibility.
34

35 **1. Introduction**

36

37 Alginates are structural biopolymers which comprise a broad family of polysaccharides found in
38 brown seaweeds (*Laminaria sp.*, *Macrocystis sp.*, *Lessonia sp.* and others) (Smidsrød & Skjak-
39 Bræk, 1990), from which they are produced industrially. In addition, bacterial (*Azotobacter*
40 *vinelandii*, *Pseudomonas aeruginosa*, and others) biosynthesis may provide alginates with more
41 defined chemical structures and physical properties or may even enable production of alginate with
42 tailor-made features (Remminghorst & Rehm, 2006). Their natural, rich and renewable sources and
43 non-toxic characteristics, accompanied by the ability of alginates to form soft hydrogels in the
44 presence of calcium ions, form the basis for a wide variety of applications in the food industry,
45 pharmacy, agriculture and environmental science (Paul, 2008). Their versatility and
46 biocompatibility provide an explanation for the wide interest shown in these molecules. For
47 example, as biomaterials, alginates can easily be formulated into a variety of soft, elastic gels,
48 fibers, foams, nanoparticles, multilayers etc. in physiological conditions that ensure the preservation
49 of cell viability and function (Andersen, Strand, Formo, Alsberg & Christensen, 2012; Goh, Heng
50 & Chan, 2012). In addition, it has been found that the alginate secreted by *Pseudomonas aeruginosa*
51 in the bronchial tract contributes to many of the problems encountered in Cystic Fibrosis (Morris &
52 Harding, 2009).

53

54 From a chemical point of view, alginate is a heteropolymeric, strictly linear (unbranched) chain
55 which consists of β -1,4-linked mannuronic acid (or its salt form) residues (M) and its C5-epimer, α -
56 L-guluronic acid (or its salt form) (G) (see, for example, Draget, Smidsrød & Skjak-Braek, 2005).
57 Alginates isolated from algae are generally of high molecular weight, typically in the range of 10^5
58 to 10^6 Da, corresponding to about 500-5000 residues per chain. Some bacterial alginates may be
59 even larger (Steigedal et al., 2008).

60

61 The properties of dilute alginate solutions reflect the flexibility and extension of the chains, which

62 are usually considered to be semiflexible. Two characteristics of these polysaccharides are essential
63 to explain their flexibility: the sequential structure of the chain (amount and distribution of G and M
64 monomers) and the polyelectrolytic nature of these molecules. This aspect can affect their
65 conformational properties (Volk, Vollmer, Schmidt, Oppermann, & Huber, 2004) because
66 intramolecular repulsion between negative charges (polyelectrolytes are macromolecules which can
67 become ionised when they are dissolved in aqueous media) add to the expansion caused by ring and
68 binding geometries. These charges along their skeleton provoke changes which can be tuned by the
69 presence of inorganic salts. High ionic strength (I) leads to a decrease in the viscosity of the solution
70 compared with salt-free conditions because when an inert salt is added the screening of the charges
71 that takes place means that the electrostatic interactions decrease and the conformation of the chain
72 becomes more compact (Pamies, Rodríguez Schmidt, López Martínez & García de la Torre, 2010).
73 Despite this simplified picture, polyelectrolytes in solution are considered among the least
74 understood systems, especially compared with neutral polymer solutions (Yethiraj, 2009).

75
76 Data sets of radius of gyration (R_g), intrinsic viscosity ($[\eta]$), sedimentation velocity (s) and even the
77 translational diffusion coefficient (D) versus molecular weight permit not only simple estimates of
78 chain conformation type (sphere rod, coil, etc.) from a power law ($P=KM^{ap}$, with $P=R_g$, $[\eta]$, s, D) or
79 “Mark–Houwink–Kuhn–Sakurada” (MHS)-type analysis but also estimates of the flexibility via the
80 chain persistence length (L_p) from more sophisticated representations. But, a survey of the literature
81 shows that the determination of persistence lengths for polysaccharides in general (Morris et al.
82 2008) and for alginates in particular (Vold, Kristiansen & Christensen 2006) is not trivial because of
83 both, experimental and theoretical-modelling-data processing difficulties.

84
85 Nowadays size exclusion chromatography (SEC) is probably the most widely used method for the
86 molecular characterization of polymers in general. SEC has the advantage that the sample is
87 chromatographically separated according to the molecular size (actually according to the

88 hydrodynamic volume, which differs markedly for different macromolecular architectures), and the
89 chromatogram gives an immediate impression of the size distribution (broad/narrow, mono- or
90 multimodal) (Mori & Barth, 1999). A breakthrough in SEC was the development of online
91 multidetectors, especially light-scattering detectors, viscometers and mass spectrometers detectors
92 (Gaborieau & Castignolles, 2011). Such SEC-multidetector systems provide the concentration
93 profiles and molecular weights, radii of gyration and intrinsic viscosity for each elution slice.

94

95 Ortega and García de la Torre (2007) developed a new combined analysis method, which forms the
96 basis of the software package, Multi-HYDFIT (Ortega & García de la Torre, 2013), a new global
97 method combining all available data sets and minimising a target (error) function, that permits a
98 more robust analysis. Using a combination of the Bohdanecky (1983) and Yamakawa and Fujii
99 (1973) representations of worm-like coils, this program gives a combined or “global” estimate of
100 the worm-like chain parameters: L_p and also the M_L (mass per unit length) and d (the chain
101 diameter).

102

103 We present a study of the influence of ionic strength on the flexibility of alginate using the SEC
104 technique. With this source of experimental data, two different data treatments are used to obtain L_p ,
105 including the HYDFIT analysis program which seems to be the best choice at present.

106

107 **2. Materials and Methods.**

108

109 **2.1. Materials.**

110

111 Sodium alginate was acquired from Sigma–Aldrich (referenced as A2158). These polymers are
112 extracted from the alga *Macrocystis pyrifera* and have an M/G ratio of 1.56 (M=61%, G=39%).

113 Dilute solutions of sodium alginate were prepared by dispersing the polymer in a previously

114 prepared NaNO₃ solution of the proper concentration for the desired ionic strength and stirring for
115 one day at room temperature. All solutions were filtered (0.22 μm filters) to remove impurities
116 before use. For this type of alginates dn/dc=0.150 (Martinsen, Skjåk-Bræk, Smidsrød, Zanetti, &
117 Paoletti, 1991).

118

119

120 **2.2 Capillary viscosimetry.**

121

122 We determined the intrinsic viscosity, $[\eta]$, by the traditional technique of capillary viscometry.

123 Solutions were prepared by isotonic dilution (Morris, Cutler, Ross-Murphy, Rees & Preece, 1981)

124 and measured using Ubbelohde viscometers (model AVS 310 from Schott Geräte). The size of the

125 capillaries was 0.53 and 0.63 mm, depending on the flow time of the solution. All the experiments

126 were carried out at 303 K and the solution flow times were in the range of 150-400 s. The intrinsic

127 viscosity was calculated by double extrapolation to a zero concentration using the classical

128 equations of Huggins (1942) and Kraemer (1938). The data were analyzed by means of VISFIT, a

129 computer program developed in our research group and available in our webpage

130 <http://leonardo.fcu.um.es/macromol>. This program simultaneously fits the data to both equations

131 providing a unique result of $[\eta]$, as detailed in López Martínez, Díaz, Ortega & García de la Torre

132 (2003).

133

134 **2.3. SEC with Online Multiangle Laser light-scattering and Viscometry.**

135

136 Measurements were carried out at 303 K. The system consisted of a solvent reservoir, on-line

137 degasser, pump, autoinjector, precolumn, and three columns (serially connected): an A6000M and

138 A4000 both from Viscotek and a PL-aquagel-OH-40-8 μm from Agilent Technologies. In some

139 experiments, the first column was changed for a TSK gel G5000 PWXL from Tosoh Bioscience.

140 The column outlet was connected to a triple detector array (TDA305 from Viscotek), to measure
141 refraction index, light scattering (two angles RALS at 90° and LALS at 7°) and viscosity (4
142 capillary differential Wheatstone bridge configuration). The integrated detectors and columns were
143 fully temperature controlled. The flow rate was 0.5 mL/min in all cases. The injection volume was
144 100 µL, and the sample concentration was adjusted to 4mg/ml for best results in the $[\eta]$ -M
145 relationship data. NaNO₃ solutions were used as mobile phases.

146
147 System, data acquisition and analysis were handled by OmniSEC software (Viscotek). This software
148 calculates M and $[\eta]$ independently and with no modeling assumptions from signals from light
149 scattering detectors and viscometer. To obtain M we have assumed the approximation $2A_2Mc \approx 0$,
150 being A_2 the second virial coefficient. This software also calculates R_g using the random coil model
151 and we found that these experimental conditions were not suitable to obtain reliable data for R_g
152 below $M=10^5$ Da. As a consequence we have not used R_g data in this work. OmniSEC calculates R_h
153 using the well known Einstein's equations assuming the molecule is a hard sphere. This is not a
154 valid assumption for our system, and in this work R_h values are only used to have an idea of the
155 evolution of the relative size of the molecules with ionic strength and no conclusions are extracted
156 from this parameter.

157
158 Data from slices were exported to a spreadsheet for further processing. In order to avoid
159 uncertainties, the data for the very low or very high M slices were not included in the analysis.

160

161 **2.4. Data treatments.**

162

163 We used two different treatments to obtain L_p , M_L and d from the experimental data. First, we
164 analyzed the $[\eta]$ -M data by basically following the analysis performed by Mendichi, Soltés, &
165 Schieronì (2003) on hyaluronan, based on the equations introduced by Bohdanecky (1983) in which

166 $(M^2/[\eta])^{1/3}$ is a linear function of $M^{1/2}$. The same procedure was used by Vold, Kristiansen &
167 Christensen (2006) to quantify the flexibility of alginates obtaining persistence length from SEC
168 data.

169
170 Secondly, the HYDFIT analysis introduced by Ortega and Garcia de la Torre (2007, 2013) was used
171 to estimate the molecular structure or parameters by performing a global weighted fit of multiple
172 samples in order to minimize a target function. The procedure was recently upgraded to include
173 more advanced computational procedures for predicting the properties of wormlike chains (Amorós,
174 Ortega & García de la Torre, 2011), and allows the simple, efficient and accurate estimation of the
175 wormlike chain parameters L_p , M_L and d . In order to reduce the number of parameters fitted
176 simultaneously, an estimate for the chain diameter d can be fixed a priori because the minimization
177 procedure is not generally sensitive to the value chosen (Patel et al., 2008). Furthermore, in cases
178 where M_L is known, the minimization procedure may yield a better defined value for L_p . In the
179 present work the hydrodynamic radius was obtained from intrinsic viscosity.

180

181 **3. Results and discussion**

182

183 **3.1. Global SEC results.**

184

185 We studied a polydisperse ($M_w/M_n=2.32$) relatively small ($M_w=107$ KDa) sample of alginate from
186 *M. pyrifera* (Table 1). The global values of $[\eta]$ (obtained from capillary viscosimetry and SEC) and
187 the hydrodynamic radius show a relatively small decrease in dimensions and intrinsic viscosity as
188 the ionic strength increases.

189

190 The agreement between M_n and M_w obtained at different ionic strength is good. Only values for
191 $I=0.4$ clearly disagree. The tendency of alginates to aggregate (or form microgels) when ionic

192 strength increases is well known (Mackie, Noy & Sellen, 1980). Figure 1 shows the weight fraction
 193 distribution for three ionic strengths. At I=0.01 and I=0.1 the weight distribution is similar, and only
 194 a nearly negligible displacement towards a higher M is observable. At I=0.4 the distribution is
 195 clearly broader, reflecting the presence of aggregates, which would explain the higher M_w than
 196 obtained in the other experiments.

197

Ionic strength	0.01	0.05	0.1	0.2	0.3	0.4	average ^b
M_n (D) $\times 10^{-3}$	44 \pm 4	46 \pm 5	47 \pm 3	43 \pm 9	48 \pm 5	49 \pm 4	46 \pm 2
M_w (D) $\times 10^{-3}$	104 \pm 4	105 \pm 5	108 \pm 9	105 \pm 7	114 \pm 4	289 \pm 40	107 \pm 4
Radius - (nm) ^c	19.9 \pm 0.3	18.5 \pm 0.5	18.2 \pm 0.4	17.8 \pm 0.6	18.2 \pm 0.3	19.7 \pm 0.6	
$[\eta]$ (cm ³ /g)	601 \pm 1	474 \pm 2	453 \pm 12	427 \pm 8	418 \pm 1	405 \pm 9	
$[\eta]$ (cm ³ /g) ^d	657	480	464	432	--	390	

198

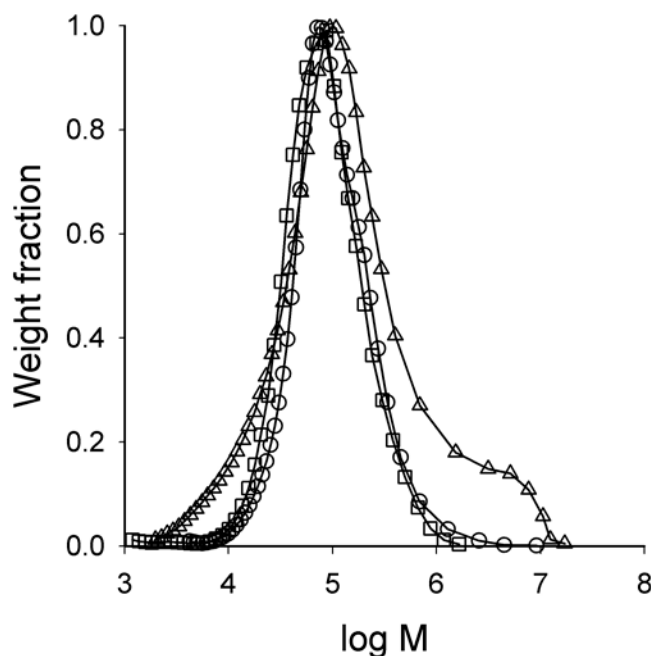
199 ^a Values and standard deviations for each ionic strength are obtained from data of 6 to 8 different injections of 2
 200 different preparations.

201 ^b Averages and standard deviations from values of M_n and M_w at different ionic strengths with values for I=0.4 not
 202 included.

203 ^c Weight average.

204 ^d From capillary viscosimetry.

205



206

207 **Figure 1.-** Weight fraction distribution of one injection for I=0.01 (□), I=0.1 (○) and I=0.4 (Δ). For better
 208 comparison, each curve has been normalized to their maximum value.

209

210 To obtain M of each slice, we have assumed $2A_2Mc \approx 0$. A_2 , the second virial coefficient, could
211 depend on ionic strength (the lower I the higher A_2) and also on molecular weight. In practice,
212 although a previous determination of A_2 should be desirable, the knowledge of A_2 in the whole
213 range of I and in the whole range of M is very difficult. For reference a value of $B=29.0 \times 10^{-4}$
214 ml.mol/g^2 was given by Wedlock et al. (2006) for a polydisperse sample of $M_w=350$ kDa from
215 *Laminaria hyperborean* at $I=0.3$. In our case, $M_w=107$ kDa. Although initial concentration
216 introduced in the column is 4mg/ml , the actual concentration of the slices is much slower reaching a
217 maximum of 0.1 mg/ml . With these data, we could estimate, on average, an underestimation around
218 5% for $I=0.3$. According with a plot shown by Horton et al. (1991) for alginate from *Laminaria*
219 *hyperborean* at $I=0.3$ and with $M_w=240$ kDa, the underestimation in our case could be even lower.
220 The worst conditions for our approximation to be valid could be when $I=0.01$. Table I shows that
221 values for M_w are independent of ionic strength in our work, except for $I=0.4$. In addition, Figure 1
222 shows that the molecular weight distribution for $I=0.01$ and $I=0.1$ are nearly identical. Figure 1 also
223 explains the main difference in M_w for $I=0.4$, showing that a significant number of aggregates have
224 appeared, probably due to the high ionic strength. Finally, Table I shows that values for IV obtained
225 from capillary viscosimetry are very similar to those obtained from SEC, what is re-assuring about
226 the use of this approximation. Anyway, this approximation must be treated carefully, because it
227 could be a limiting factor, mainly when we work with longer chains and at low ionic strengths.

228

229 **3.2. Results for infinite ionic strengths.**

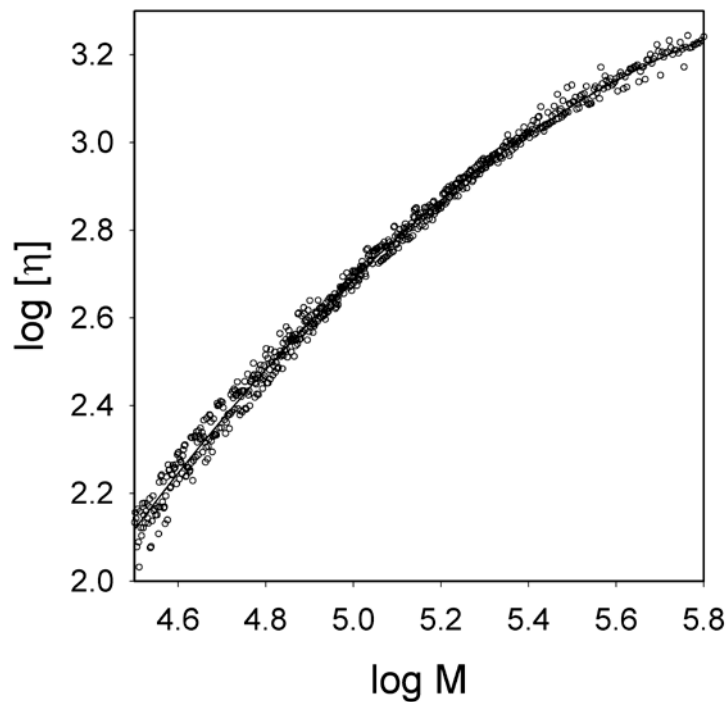
230

231 To obtain results for infinite ionic strengths it is necessary to find a reliable extrapolation method
232 using the Pal-Hermans equation (Pals & Hermans, 1950, 1952)

$$233 \quad [\eta] = [\eta]_{\infty} + SI^{-1/2}$$

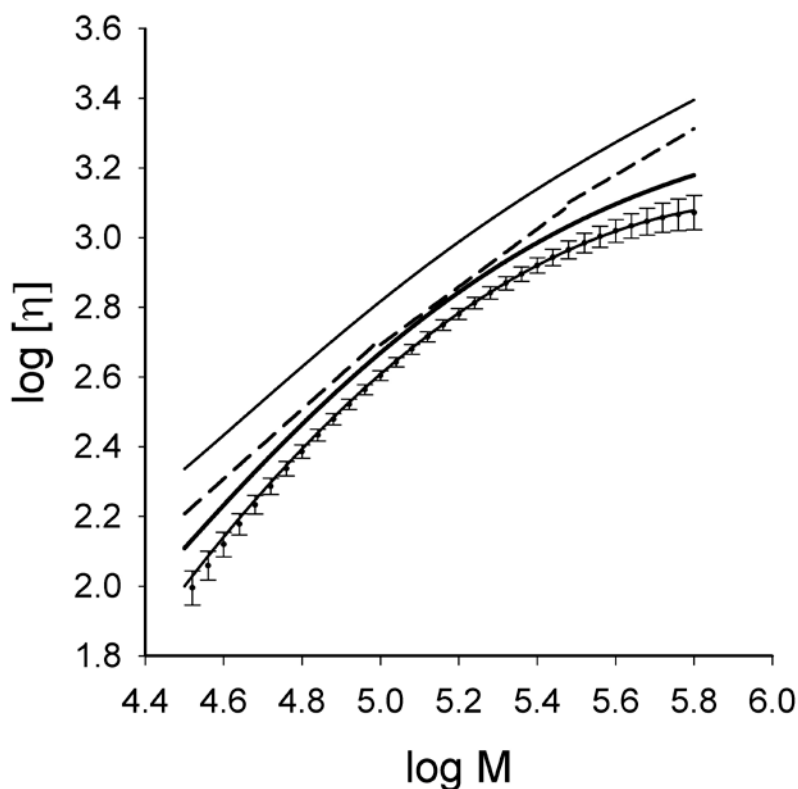
234 To handle the large amount of experimental data, a series of steps were followed. The first one
235 involved fitting data to an appropriate curve. The MHS power law relationships between M and $[\eta]$

236 is usually accepted, but, according to our data, in a range from $\log M=4.5$ to $\log M=5.8$ there is a
237 curved instead of a linear dependence (as expected from MHS law). This curvature was already
238 observed by Vold et al. (2006) although their option was to divide the data into three ranges of
239 molecular weights (Vold, Kristiansen & Christensen, 2007). We decided to fit our data to a second
240 order polynomial equation, with good results (not shown). An illustrative example is shown in
241 Figure 2 for $I=0.1$, including the fitting equation $\log [\eta] = -13.027 + 4.502 \log M - 0.35549 (\log M)^2$.



242
243 **Figure 2.-** Illustration of the fitting the experimental data to a second order polynomial equation in the $\log[\eta]$
244 versus $\log M$ plot. The data shown are for $I=0.1$. The fitting equation is: $\log [\eta] = -13.027 + 4.502 \log M -$
245 $0.35549 (\log M)^2$. Two different alginate solutions were prepared with 8 injections for each. For better
246 visualization, only 1 of every 75 data is shown in the plot.

247



248

249 **Figure 3.-** Log plot of the $[\eta]$ - M relationship for various ionic strengths obtained after taking into account all
 250 experimental results (solid lines). From lower to higher: $I=\infty$, $I=0.17$ and $I=0.01$. $[\eta]_{\infty}$ was obtained as the
 251 average of all ionic strengths according to equation 4. Error bars are included. $[\eta]_{0.01}$ and $[\eta]_{0.17}$ were
 252 obtained from $[\eta]_{\infty}$ and equation 4. Results from Vold et al. (2007) for $I=0.17$ are included for comparison
 253 (dashed line). $[\eta]$ is in mL/g; M in Da.

254

255 The second step was to obtain S as a function of M . We observed that a second order polynomial
 256 described the S dependence on M very well ($S = 2.351 + 2.333 \times 10^{-4} M - 6.244 \times 10^{-11} M^2$). Once this
 257 dependence is known $[\eta]_{\infty}$ could be readily obtained for each value of M by applying the Pal-
 258 Hermans equation from data of any ionic strength. We decided to take into account all available
 259 data and we obtained the averaged value of $[\eta]_{\infty}$ for each M . Later, data were fitted to a second
 260 order polynomial equation as shown in Figure 3.

261

262 Another option, which was used by Vold. et al. (2006, 2007), would be to use the empirical B
 263 parameter (Smidsrød & Haug, 1971), assuming that is molecular weight-independent. B can be

264 obtained from the Pals-Hersman equation defining $S = B[\eta]_{0.1}^{\nu}$, with $\nu=1.3$ (Dentini, Rinaldi, Risica,
265 Barbetta & Skjåk-Bræk, 2005). We obtained B from the SEC quasi-monodisperse fractions and
266 since our results showed a significant dependence on M this option was discarded.

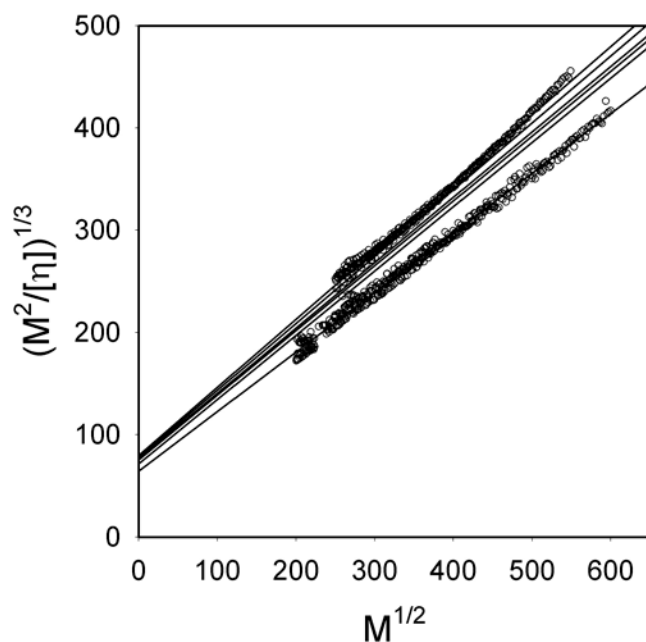
267

268 3.3. Obtaining persistence length.

269

270 We applied the HYDFIT analysis and Bohdanecky's equations to the M and $[\eta]$ data sets obtained
271 from SEC elution slices to obtain persistence length of alginate molecules.

272



273

274 **Figure 4.-** SEC results in the form of Bohdanecky's equations. Only some experimental data for two values
275 of I (I=0.01, lower in plot and I=0.4, higher in plot) are shown for clarity. For the same reason, only 1 of
276 every 50 experimental points is plotted. Lines are obtained from linear regression in slightly different
277 molecular weight ranges (see text) and correspond, from lower to higher, to I=0.01, 0.05, 0.1, 0.2, 0.3 and
278 0.4. Fitting results are shown in Table 2.

279

280 When using Bohdanecky's equations (Figure 4), linear regressions of $(M^2/[eta])^{1/3}$ versus $M^{1/2}$ for
281 each ionic strength give the intercepts and slopes and therefore L_p , which is sensitive to slight

282 differences in these parameters. In the data fitting procedure it is desirable to use the widest possible
 283 range of molecular weights, although we found that the range of linear behavior decreased
 284 significantly with increasing I. To increase the consistency of our data between 8 and 15 different
 285 injections of two different preparations were used. In addition we used two different (although
 286 similar) sets of chromatographic columns. Accordingly, ranges run from M=40000 to 360000 for
 287 I=0.01 to M=60000 to 300000 for I=0.4. Values for the fitting parameter R^2 are always over 0.99,
 288 what it is reasonably taking into account that the actual number of data for each fitting is several
 289 thousands. Providing the partial hydrodynamic volume (\bar{v}) is known, the hydrodynamic diameter
 290 can be calculated and then M_L and L_p . We used $\bar{v}=0.54\pm 0.06$ mL/g (Martin, Cook & Winkler, 1956).
 291 Values for M_L and L_p are shown in Table 2.

292

293 According to this treatment, the persistence length is around 12.0 nm and does not change
 294 significantly with the ionic strength of the solution (at least in the range we used) and the changes
 295 are produced in M_L (from 323 to 420 nm⁻¹) and, to a less significant extent, in the diameter of the
 296 chain.

297

Table 2.- Values for M_L and L_p for different ionic strengths obtained from Bohdanecky's equations and the HYDFIT procedure.

I	HYDFIT				Bohdanecky		
	d (nm)	M_L (Da/nm)	L_p (nm)	Δ^2 ($\times 10^2$)	d (nm)	M_L (Da/nm)	L_p (nm)
0.01	0.64	321	11.7	2.13	0.61	323	12.1
0.05	0.60	384	13.2	2.50	0.65	365	11.8
0.1	0.66	383	12.1	1.79	0.66	384	12.2
0.2	0.66	383	11.5	1.42	0.67	391	12.2
0.3	0.66	420	13.1	2.30	0.68	401	11.9
0.4	0.80	413	11.3	1.23	0.69	412	11.8
∞	0.71	464	14.2	1.26	0.70	426	12.0
Average			12 \pm 1				12.0 \pm 0.2
0.01			19.3	3.21			15.5
0.05			14.7	2.45			13.2

0.1	0.69	410	13.7	1.81	0.69	410	13.1
0.2			13.2	1.51			12.8
0.3			12.1	2.45			12.2
0.4			11.6	1.36			11.8
∞			11.1	1.35			11.5

From data with M in g/mol and $[\eta]$ in cm³/g

298
299
300

301 The M_L value can be obtained theoretically from the chemical structure, according to the relation

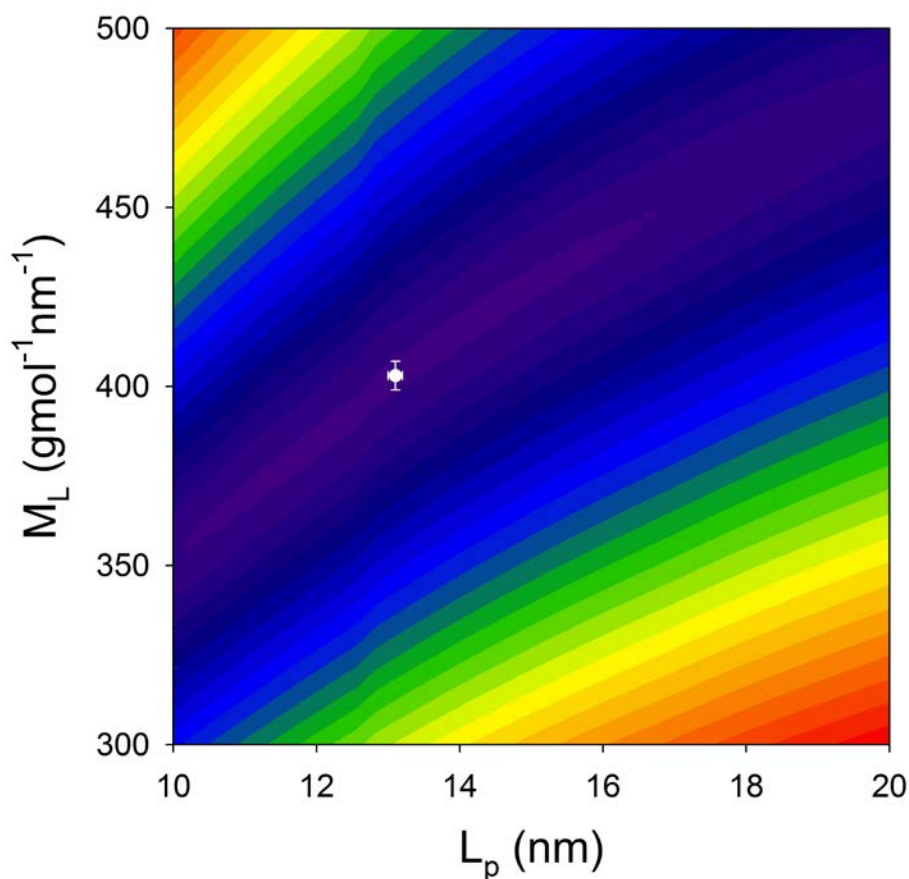
302
$$M_L = \frac{F_G M_{0,G} + (1 - F_G) M_{0,M}}{F_G b_G + (1 - F_G) b_M}$$

303 Where F_G is the fraction of L-guluronic acid residues, and $M_{0,G}$ and $M_{0,M}$ are the molar masses of
304 the two monomers (both 198 g/mol for the Na⁺ salt). The bond distances are 0.515 nm (b_M) and
305 0.435 nm (b_G), respectively, with the first referring to the M in the ⁴C₁ form, and G in the ¹C₄ form
306 (Smidsrød, Glover & Whittington, 1973). In our case, $M_L=410$ Da/nm, and with this constant value,
307 L_p values also can be obtained as shown in Table 2.

308

309 The use of HYDFIT is simpler because it is simply necessary to introduce a reasonably wide range
310 of d, M_L and L_p . After introducing the same sets of input data as used with Bohdanecky's equations
311 we obtained the results shown in Tables 2 after optimizing a target function Δ (Figure 5).

312



313

314 **Figure 5.-** Contour plot for HYDFIT target function Δ for $I=0.1$ with $d=0.69$ nm. In this representation the
 315 values of Δ are presented by the full colour spectrum, from blue ($\Delta=0.05$) to red ($\Delta=25$). The minima (with
 316 error bars is also plotted (white).

317

318 **3.5. Flexibility of alginate**

319

320 We have used the wormlike model to quantify the flexibility of alginate molecules. Both data
 321 treatments (Bohdanecky's equations and the HYDFIT program) give very similar results for
 322 persistence length (Tables 2). However, it must be concluded that this model is not sufficient to
 323 describe the $[\eta]$ - M relationship over a wide range of M (see Figure 4 as an illustration), an
 324 observation that becomes relevant as I increases. Vold et al. (2006) also found that the wormlike
 325 chain may not be the most suitable model when the flexibility of alginate molecules increases, as
 326 occurred in their work in the absence of excluded volume effects for the most oxidized alginate
 327 samples.

328

329 In the analysis of our experimental data two different assumptions led to different conclusions.
330 First, as frequently done in previous studies, M_L was assumed to be constant. The expected increase
331 in persistence length as I decreased was indeed observed. For polyelectrolytes, persistence length
332 has usually been described as the sum of two contributions (Odijk, 1977; Skolnick & Fixman,
333 1977): the intrinsic contribution, due to the chemical structure of the chain in the absence of
334 intramolecular electrostatic interactions, and the electrostatic contribution due to the
335 polyelectrolytic nature of the molecule. In our case, the L_p values of 11.5 and 11.1 nm were
336 obtained for $I=\infty$ (averaged, 11.3 nm). The value of this intrinsic contribution is only slightly
337 smaller than that (12.5 nm) obtained by Zhang H., Wang H., Wang J., Guo R., Zhang Q. (2001) for
338 longer alginates from *L. nigrescens* following the method suggested by Odijk (1977). This value is
339 also close to that (12 nm) proposed in the more recent paper of Vold et al. (2006) for *L.*
340 *hyperborean*. We find that the electrostatic contribution to persistence length at $I=0.01$ would be
341 around 6 nm (see Table 2). This value is higher than the one obtained by Zhang et al. (2001) (3.2
342 nm). At $I=0.2$, we find the electrostatic contribution to L_p to be 1.7 nm, while, for $I=0.17$, Vold et al.
343 (2006) obtained 3.3 ± 0.3 nm ($M_L=440\pm 10$ Da/nm) and 2.7 nm ($M_L=424$ Da/nm). Although with
344 some differences, all these values mean that the intrinsic contribution to flexibility of the structure
345 of the chain would be higher than the effect of the external salt in this range of ionic strengths.

346

347 Another possibility was not to impose any restriction for calculating M_L or L_p (indeed, with the
348 large set of M -dependent $[\eta]$, both parameters can be safely adjusted simultaneously with the
349 Bohdanecky method or the HYDFIT program). In this case (see Table 2) a new explanation (we
350 have not found it in the literature) of the effect of the presence of inorganic ions in solutions of
351 alginates (and perhaps of other polyelectrolytes) can be deduced: the increase in ionic strength
352 would produce a decrease in contour length and, as a consequence, in M_L (also a small change in
353 diameter) and small changes in L_p which value would be around 12 nm.

354

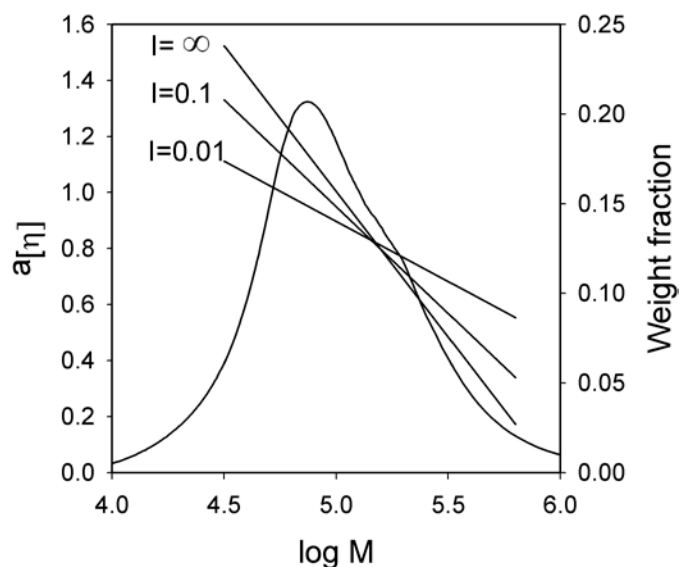
355 HYDFIT allows Δ to be obtained, that is, a parameter to evaluate the goodness of the found target
356 function. This parameter is always slightly smaller (better) for the treatment in which M_L is not
357 fixed as opposed to when M_L is taken to be 410 Da/nm, which supports the second option.

358

359 A different and complementary approach is the analysis of the MHS exponent obtained from
360 experimental data. All our SEC results show that an increase in I leads to a decrease in $[\eta]$ for
361 molecules of the same M . In addition, once the $[\eta]_{\infty}$ - M relationship was obtained taking into
362 account all available data, it is a straightforward task to calculate the $[\eta]$ - M relationship for any
363 ionic strength (Figure 3). Such relationships would be valid in the $I=0.01$ to ∞ and $\log M=4.5$ to 5.8
364 ranges.

365

366 Moreover, these relationships are well described by a second order polynomial, and, as a
367 consequence, if $\log[\eta]=a_0+a_1(\log M)+a_2(\log M)^2$, a local MHS exponent dependent on M can be
368 defined as $a_{[\eta]}(M)=a_1+2a_2(\log M)$ (see Figure 6).



369

370 **Figure 6.**-Local Mark-Howking-Sakurada $a_{[\eta]}$ as a function of M for three different ionic strengths (straight
371 lines). Weight fraction distribution of one sample ($I=0.1$) is also included as a reference (curve). Equations
372 are as follows: $I=\infty$, $a_{[\eta]}= 6.204 - 1.040 \log M$; $I=0.1$, $a_{[\eta]}= 4.759 - 0.762 \log M$; $I=0.01$, $a_{[\eta]}= 3.047 -$

373 0.430 log M.

374

375 Vold et al. (2006) found no qualitative differences in the dependence of M and $[\eta]$ between $I=0.17$
376 and $I=1$. Our results (Figure 6) point to a range of M in which no or very slight differences in $a_{[\eta]}$
377 are observed for different values of I, even if significantly different flexibilities (or conformations)
378 are observed at higher M. This might explain the results of the mentioned paper.

379

380 $a_{[\eta]}$ is usually used as a semi-quantitative measure of the rigidity or the expansion of the molecule
381 (Harding, Abdelhameed & Morris, 2011), assigning more extended conformations to higher values
382 of $a_{[\eta]}$. Therefore, Figures 3 and 6 show that, for low M values, molecules become more rigid (more
383 extended) when I increases. The same plots show that in the high M range molecules become more
384 flexible (less extended) when I increases. This apparent contradiction suggests a complex influence
385 of the inorganic ions present in the solution, including the tendency of alginate molecules to
386 aggregate when the number of ions present in the solution grows.

387

388 A plausible explanation for all these observations could be as follows. When I is low, the alginate
389 molecules are well described by the wormlike model, but when concentration of inorganic ions
390 grows two different effects could appear. On the one hand, short range interactions or screening
391 between COO^- groups of contiguous or very close G and M monomers could shorten the contour
392 length of the molecules as the salt concentration increases, giving rise to shorter (lower $[\eta]$) and
393 more rigid molecules. At the same time long range interactions or screening between charges more
394 separated in the chain could allow more folded conformations, and, for higher ionic strengths,
395 would also allow aggregation among different molecules. These two effects would therefore be
396 opposite and in competition and will depend on ionic strength. For shorter molecules, the first factor
397 would be more important while the second would increase strongly with chain length.

398

399 **3.6. Differences found in the literature.**

400

401 A survey of the literature finds variable results in the characterization of flexibility of alginates.

402 Vold et al. (2006) attributed the differences to sample quality and purification, as well as data

403 processing. They also mention that slightly different analysis conditions (temperature, ionic

404 strength, shear rate) were used, although these parameters generally fell within ranges that would be

405 expected to play only a minor, or even negligible, role. Another source of difficulties could be the

406 chemical composition of alginates. Indeed, experimental and theoretical studies have concluded that

407 polyguluronate blocks (GG) are stiffer than polymanuronate blocks (MM), while heteropolymeric

408 blocks (MG) are the most flexible (Mackie, Noy, & Sellen, 1980; Smidsrød, 1970), although some

409 recent studies have found no such dependence on the composition (Vold, Kristiansen &

410 Christensen, 2006; Storz et al. 2009; Josef & Bianco-Pelet, 2012). Our results show that

411 polydispersity could add some complexity to the interpretation of the experimental results.

412

413 Although we found that the MHS equation is not a good representation of the experimental results

414 for wide ranges of molecular weights, historically this form it has been used. In the same way as

415 Vold et al. (2006) did, the ranges of M can be fractionated to perform an approximate fitting to a

416 straight line of the relationship $\log[\eta]$ - $\log M$. For illustrative purposes, we divided our results, for

417 $I=0.1$, in four different M ranges obtaining $a_{[\eta]}=1.13$ for $M=40-80$ kDa, $a_{[\eta]}=0.91$ for $M=80-160$

418 kDa, $a_{[\eta]}=0.68$ for $M=160-320$ kDa and $a_{[\eta]}=0.47$ for $M=320-630$ kDa. The second range (see

419 Figure 1) covers a large fraction of the total mass of the sample and is where the maximum RI

420 signal is found in our SEC measurements. It is also a region in which the signal to noise ratio for all

421 detectors is good and the one that includes the value of M_w . The MHS representation in this range

422 of molecular weights gives very similar results to those of Martinsen et al. (1991) for alginates from

423 the same source. But, if molecules of lower M are included, $a_{[\eta]}$ would increase, while the contrary

424 would occur if molecules of higher M were included.

425

426 For comparison, we have included in Figure 3 the results obtained by Vold et al. (2007). The
427 coincidence between both works is very good in the range between 100 and 200 kDa. But
428 significant quantitative differences appear at low and high M. These differences could be somehow
429 coherent with the results shown in our Figure 6, which shows that differences in conformations
430 (although this figure compares the same alginate at different ionic strengths) are going to be small
431 in a molecular range around 150 kDa. We think that this is an interesting topic of study in the future
432 because we have not found a definitive explanation. One reason to explain the differences could be
433 the nature of the alginate sample because the composition in terms of monomers is rather different
434 (60% M versus 30% M, approximately) but, as suggested by different authors (Vold, Kristiansen &
435 Christensen, 2006; Storz et al. 2009; Josef & Bianco-Pelet, 2012), we think that this is unlikely.
436 Another factor to explain the difference could be the procedure to obtain molecular weight. Vold et
437 al. (2006, 2007) used a multiangle light scattering detector and data processing according to the
438 Zimm formalism while we have used a two angle light scattering detector as explained before. But,
439 as we have also discussed above, we think that it is unlikely that differences are due to this factor.
440 In addition, our Figure 3 show the curves obtained after taking into account all the results that we
441 have measured at different ionic strengths, being all of them coherent in absolute and relative terms.

442

443 According to our results the influence of the chain length distribution and the ionic strength on the
444 $[\eta]$ of a polydisperse sample is very important and quite complex. Indeed, as mentioned by Vold et
445 al. (2006, 2007), the $a_{[\eta]}$ parameter of the MHS equation decreases significantly at increasing
446 molecular weights even for molecules with the same L_p . Moreover, for high I, not only individual
447 molecules, but also aggregates of different sizes could be present, which might make $a_{[\eta]}$ to be even
448 smaller. Finally, we remind that when simultaneously analyzing a mixture of different chain lengths,
449 as in a polydisperse system, only an average is observed. As a consequence, the analysis of the
450 MHS equations found in the literature would need to take into account these factors.

451

452 As an additional factor to understand the differences found in the literature, we must remind that the
453 determination of absolute values of M is strongly affected by dn/dc , and there are significant
454 differences between the values presented in different works from 0.168 (Rinaudo, 2008) or 0.165
455 (Theisen, Johann, Deacon & Harding, 2000) to the lower value of 0.150 (Martinsen et al., 1991).

456

457 **4- Conclusions**

458

459 SEC experiments of polydisperse alginate from *M. pyrifera* are an important source of information,
460 but the quality and statistical robustness of data for a wide range of molecular weight are essential
461 to reach reliable conclusions.

462

463 According to our results, the MHS power law does not provide a good description of the $[\eta]$ - M
464 relationship for wide ranges of molecular weights of monodisperse fractions. This could explain
465 some of the differences in the interpretation of the $[\eta]$ - M data found in the literature.

466

467 The quantitative determination of the parameters for these polyelectrolytic chains using the
468 wormlike model shows two different scenarios. The first one appears if we assume that M_L (and so
469 contour length and diameter) do not change by the effect of I . In this case the “classical” flexibility
470 mechanism of L_p decreasing with ionic strength would be adequate. The intrinsic component of the
471 persistence length would be around 11.3 nm and the electrostatic one would be around 6 nm when
472 $I=0.01$. These values are close to previous results from literature. In the second scenario, we do not
473 impose any restriction on the calculation of M_L or L_p in the wormlike model. Then a new
474 description of the effect of inorganic ions on alginates (and perhaps other polyelectrolytes) could be
475 offered. In this case, the decrease in $[\eta]$ with increasing ionic strength would be due to by a decrease
476 in contour length and hence in M_L . Any changes in the value of L_p which, in our case, is ± 12 nm

477 would be slight. Although further investigation is needed, this option offers a new way to
478 understand the complex mechanism of flexibility in polyelectrolytes.

479

480 Both of the two former options use the wormlike model, but this in itself is not enough to explain
481 flexibility over the whole range of chain lengths, particularly at higher values of ionic strengths.

482 One plausible explanation could be the combination of short-range and long-range screening effects
483 of inorganic ions in the solutions. These effects would allow the formation of aggregates when the
484 salt concentration is high enough.

485

486 Finally, it should be emphasized that additional information on R_g and other hydrodynamic
487 properties would be desirable. The use of previously fractionated samples should reduce some of
488 the difficulties provoked by polydispersity. Also the computer simulation of adequate models would
489 help to improve our knowledge of the flexibility mechanisms of polyelectrolytes in general and
490 alginates in particular.

491

492

493 **Acknowledgments**

494

495 This work was performed within a Grupo de Excelencia de la Región de Murcia (grant
496 04531/GERM/06). Support also provided by grant CTQ-2012-33717 from Ministerio de Economía
497 y Competitividad including FEDER funds. A.O.R. acknowledges a postdoctoral fellowship from
498 Fundación CajaMurcia, and A.I.D.P. is recipient of a predoctoral fellowship from MEC. We are
499 grateful to Ricardo Rodríguez Schmitz (Department of Physical Chemistry, University of Murcia),
500 Alejandro Torrecillas Sánchez and Vanessa Cánovas Cánovas (CAID, University of Murcia) for
501 their help.

502 **References**

- 503 Amorós, D., Ortega, A., & García de la Torre, J. (2011). Hydrodynamic properties of wormlike
504 macromolecules: Monte Carlo simulation and global analysis of experimental data.
505 *Macromolecules*, *44*, 5788-5797.
- 506 Andersen, T., Strand, B. L., Formo, K., Alsberg, E., & Christensen, B.E. (2012). Alginates as
507 biomaterials in tissue engineering. *Carbohydr. Chem.*, *37*, 227-258.
- 508 Bohdanecky, M. (1983). New method for estimating the parameters of the wormlike chain model
509 from the intrinsic viscosity of stiff-chain polymers. *Macromolecules*, *16*, 1483–1492.
- 510 Dentini, M., Rinaldi, G., Risica, D., Barbetta, A., & Skjåk-Bræk, G. (2005). Comparative studies on
511 solution characteristics of mannuronanepimerized by C-5 epimerases. *Carbohydrate*
512 *Polymers*, *59*, 489-499.
- 513 Draget, K.I., Smidsrød, O., & Skjåk-Braek G. (2005). Alginates from algae. In A. SteinBüchel, &
514 S.K. Rhee. (Eds.), *Polysaccharides and Polyamides in the Food Industry. Properties,*
515 *Production and Patents* (pp. 1-30). Weinheim: Wiley-VCH Verlag GmbH & Co. KgaA.
- 516 Gaborieau, M., & Castignolles, P. (2011). Size-exclusion chromatography (SEC) of branched
517 polymers and polysaccharides. *Anal. Bioanal. Chem.*, *399*, 1413-1423.
- 518 Goh, C.H., Heng, P.W.S., & Chan, L.W. (2012). Alginates as a useful natural polymer for
519 microencapsulation and therapeutic applications. *Carbohydrate Polymers*, *88*, 1-12.
- 520 Harding, S. E., Abdelhameed, A. S., & Morris, G. A. (2011). On the hydrodynamic analysis of
521 conformation in mixed biopolymer systems. *Polym. Int.*, *60*, 2–8.
- 522 Horton, J.C., Harding, S.E., Mitchell, J.R. & D.F.Morton-Holmes, D.F. (1991). Thermodynamic
523 non-ideality of dilute solutions of sodium alginate studied by sedimentation equilibrium
524 ultracentrifugation. *Food Hydrocolloids*, *5*, 125-127.
- 525 Huggins, M. L. (1942). The viscosity of dilute solutions of long-chain molecules. IV. Dependence
526 on concentration. *Journal of the American Chemical Society*, *64*, 2716–2718.

- 527 Josef, E., & Bianco-Pelet, H. (2012). Conformation of a natural polyelectrolyte in semidilute
528 solutions with no added salt. *Soft Matter*, 8, 9156-9165.
- 529 Kraemer, E. O. (1938). Molecular weights of celluloses. *Industrial and Engineering Chemistry*, 30,
530 1200–1203.
- 531 López Martínez, M. C., Díaz, F. G., Ortega, A., & García de la Torre, J. (2003). Multiple linear
532 least-squares fits with a common intercept. Application to the determination of intrinsic
533 viscosity of macromolecules in solution. *Journal of Chemical Education*, 80, 1036–1038.
- 534 Mackie, W., Noy, R., & Sellen, D. B. (1980). Solution properties of sodium alginate. *Biopolymers*,
535 19, 1839-1860.
- 536 Martin, W. G., Cook, W. H., & Winkler, C. A. (1956). The determination of partial specific volumes
537 by differential sedimentation. *Can. J. Chem.*, 34, 809-814.
- 538 Martinsen, M., Skjåk-Bræk, G., Smidsrød, O., Zanetti, F., & Paoletti, S. (1991). Comparison of
539 different methods for determination of molecular weight distribution of alginates.
540 *Carbohydrate Polymers*, 15, 171–193.
- 541 Mendichi, R., Soltés, L., & Schieroni, A. G. (2003). Evaluation of radius of gyration and intrinsic
542 viscosity molar mass dependence and stiffness of hyaluronan. *Biomacromolecules*, 4, 1805.
- 543 Mori, S., & Barth H.G. (1999). *Size Exclusion Chromatography*. Berlin: Springer-Verlag.
- 544 Morris, E. R., Cutler, A. N., Ross-Murphy, S. B., Rees, D. A., & Preece, J. (1981). Concentration
545 and shear rate dependence of viscosity in random coil polysaccharide solutions.
546 *Carbohydrate Polymers*, 1, 5–21.
- 547 Morris, G. A., & Harding S.E. (2009). Polysaccharides, microbial. In *Encyclopedia of Microbiology*
548 (pp. 482-494). (3rd ed). M. Schaechter (Ed.). Amsterdam: Elsevier.
- 549 Morris, G.A., Patel, T.R., Picout, D.R., Ross-Murphy, S.B., Ortega, A., García de la Torre, J., &
550 Harding, S.E. (2008). Global hydrodynamic analysis of the molecular flexibility of
551 galactomannans. *Carbohydrate Polymers*, 72, 356–360.
- 552 Odijk, T.J.(1977). Polyelectrolytes near the rod limit. *Journal of Polymer Science Polymer Physics*

- 553 *Edition, 15, 477-483.*
- 554 Ortega, A., & García de la Torre J. (2007). Equivalent radii and ratios of radii from solution
555 properties as indicators of macromolecular conformation, shape and flexibility.
556 *Biomacromolecules, 8, 2464-2475.*
- 557 Ortega, A., & García de la Torre, J. (2013). HYDFIT and Related Packages for Linear Molecules. In
558 *Encyclopedia of Biophysics, Vol. 2* (pp. 998-1002). G. Roberts (Ed.). Berlin-Heidelberg:
559 Springer-Verlag.
- 560 Pals, D. T. F., & Hermans, J. J. (1950). New method for deriving the intrinsic viscosity of
561 polyelectrolytes. *Journal of Polymer Science Polymer Physics Edition, 6, 733-734.*
- 562 Pals, D. T. F., & Hermans, J. J. (1952). Sodium salt of pectin and of carboxy methyl cellulose in
563 aqueous sodium chloride. *Recueil des Travaux Chimiques des Pays-Bas-Journal of the*
564 *Royal Netherlands Chemical Society, 71, 458-467.*
- 565 Pamies, R., Rodríguez Schmidt, R., López Martínez, M.C., & García de la Torre, J. (2010). The
566 influence of mono and divalent cations on dilute and non-dilute aqueous solutions of
567 sodium alginates. *Carbohydrate Polymers, 80, 248-253.*
- 568 Patel, T. R., Morris, G. A., Garcia de la Torre, J., Ortega, A., Mischnick, P., & Harding, S.E. (2008).
569 Molecular flexibility of methylcelluloses of differing degree of substitution by combined
570 sedimentation and viscosity analysis. *Macromol. Biosci., 8, 1108-1115.*
- 571 Paul, W. (2008). *Natural bioresorbable polymers*. Cambridge: Woodhead Publishing Ltd.
- 572 Remminghorst, U., & Rehm, B.H.A. (2006). Bacterial alginates: from biosynthesis to applications.
573 *Biotechnol. Lett., 28, 1701-12.*
- 574 Rinaudo, M. (2008). Main properties and current applications of some polysaccharides as
575 biomaterials. *Polym. Int., 57, 397-430.*
- 576 Skolnick, J., & Fixman, M. (1977). Electrostatic persistence length of a wormlike polyelectrolyte.
577 *Macromolecules, 10, 944-948.*
- 578 Smidsrød, O. (1970). Solution properties of alginate. *Carbohydrate Research, 13, 359-372.*

- 579 Smidsrød, O., Glover, R. M., & Whittington, S. G. (1973). The relative extension of alginates
580 having different chemical composition. *Carbohydr. Res.*, 27, 107-118.
- 581 Smidsrød, O., & Haug, A. (1971). Estimation of the Relative Stiffness of the Molecular Chain in
582 Polyelectrolytes From Measurements of Viscosity at Different Ionic Strengths.
583 *Biopolymers*, 10, 1213-1227.
- 584 Smidsrød, O., & Skjak-Bræk G. (1990). Alginate as immobilization matrix for cells. *Trend.*
585 *Biotechnol.*, 8, 71-78.
- 586 Steigedal, M., Sletta, H., Moreno, S., Maerk, M., Christensen, B.E., Bjerkan, T., Ellingsen, T.E.,
587 Espin, G., Ertesvag, H., & Valla, S. (2008). The *Azotobacter vinelandii* AlgE mannuronan
588 C-5-epimerase family is essential for the *in vivo* control of alginate monomer composition
589 and for functional cyst formation. *Environmental Microbiology*, 10, 1760-1770.
- 590 Storz, H., Müller, K. J., Ehrhart, F., Gómez, I., Shirley, S. G., Gessner, P., Zimmermann, G.,
591 Weyand, E., Sukhorukov, V. L., Forst, T., Weber, M. M., Zimmermann, H., Kulicke, W-M.,
592 & Zimmermann, U. (2009). Physicochemical features of ultra-high viscosity alginates.
593 *Carbohydrates Research*, 344, 985-995.
- 594 Theisen, C., Johann, C., Deacon, M.P., & Harding, S.E. (2000). *Refractive Increment Data Book for*
595 *Polymer and Biomolecular Scientists*. Nottingham: Nottingham University Press.
- 596 Vold, I.M., Kristiansen, K.A., & Christensen, B.E. (2006). A study of the chain stiffness and
597 extension of alginates, in vitro epimerized alginates, and periodate-oxidized alginates using
598 size-exclusion chromatography combined with light scattering and viscosity detectors.
599 *Biomacromolecules*, 7, 2136-2146.
- 600 Vold, I. M., Kristiansen, K. A., & Christensen, B. E. (2007). Additions and corrections: A study of
601 the chain stiffness and extension of alginates, in vitro epimerized alginates, and periodate-
602 oxidized alginates using size-exclusion chromatography combined with light scattering and
603 viscosity detectors. *Biomacromolecules*, 8, 2627.
- 604 Volk, N., Vollmer, D., Schmidt, M., Oppermann, W., & Huber, K. (2004). *Polyelectrolytes with*

605 *defined molecular architecture (Vol. 166)*. Berlin, London: Springer. pp. 29-65.

606 Wedlock, D.I., Baruddin, B.A. & Phillips, G.O. (1986). Comparison of molecular weight
607 determination of sodium alginate by sedimentation-diffusion and light scattering. *Int. J.*
608 *Biol. Macromol.*, 8, 57-61.

609 Yamakawa, H., & Fujii, M. (1973). Translational friction coefficient of wormlike chains.
610 *Macromolecules*, 6, 407–415.

611 Yethiraj, A. (2009). Liquid state theory of polyelectrolyte solutions. *Journal of Physical Chemistry*
612 *B*, 113, 1539–1551.

613 Zhang, H., Wang, H., Wang, J., Guo, R., & Zhang, Q. (2001). The effect of ionic strength on the
614 viscosity of sodium alginate solution. *Polymers for Advanced Technologies*, 12, 740–745.

615

616